

# First-Principles Study of Bonding-Driven Selectivity in CO<sub>2</sub> Electroreduction on Metal–Nitrogen–Carbon Catalysts

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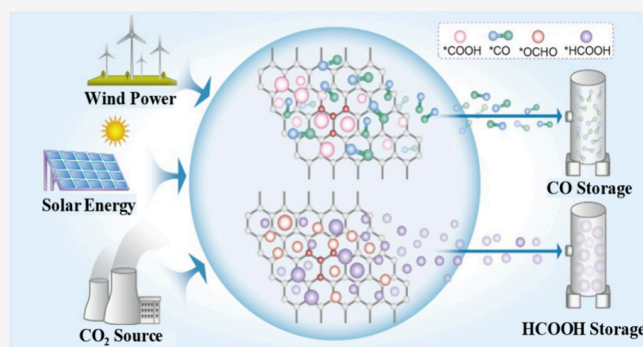


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**ABSTRACT:** Single-atom catalysts (SACs) have emerged as promising candidates for the carbon dioxide reduction reaction (CO<sub>2</sub>RR), and the design of new SACs is of great significance. This study introduces a series of transition metal SACs anchored on nitrogen-doped single-layer graphene (denoted as TM-C<sub>2</sub>N, with TM representing Fe, Ni, Cu, Pd, Ag, and Sn), designed for the selective conversion of CO<sub>2</sub> to CO or formic acid. Utilizing first-principles computational approaches, the structural integrity, CO<sub>2</sub> adsorption, and activation dynamics of these catalysts have been systematically investigated. Our findings reveal that the TM-C<sub>2</sub>N catalysts not only manifest exceptional structural stability but also exhibit superior CO<sub>2</sub> adsorption and activation capabilities, coupled with an effective suppression of the competing hydrogen evolution reaction (HER). Gibbs free energy analyses have delineated distinct reaction pathways leading to HCOOH and CO formation on TM-C<sub>2</sub>N. Notably, Ni-C<sub>2</sub>N stands out as the most active catalysts, as evidenced by their favorable limiting potentials. The role of bonding interactions in elucidating the gas–solid interface adsorption mechanism is also highlighted. These insights offer valuable theoretical guidance for the fine-tuning of C<sub>2</sub>N-based catalysts in experimental settings and have broad implications for the development of efficient transition metal SACs for CO<sub>2</sub> reduction.



## INTRODUCTION

With the development of industry and economy, more and more fossil fuels have been consumed, and the continuous intensification of CO<sub>2</sub> emissions has led to abnormal changes in global climate and a general warming trend.<sup>1</sup> Therefore, how to reduce the impact of excessive CO<sub>2</sub> on human production and life has become a major issue. Although CO<sub>2</sub> is a greenhouse gas, a large number of high concentrations of CO<sub>2</sub> can also be regarded as a rich carbon source, so the efficient conversion and effective use of CO<sub>2</sub> has become a crucial step to solve the greenhouse effect and achieve carbon neutrality.<sup>2–4</sup> CO<sub>2</sub> can be chemically transformed into valuable products such as methanol, methane, and formic acid through catalytic methods like hydrogenation and advanced conversion technologies. Formic acid, specifically, is extensively utilized as a feedstock in chemical manufacturing, pharmaceutical synthesis, and food processing industries. Recent advancements in CO<sub>2</sub> conversion have introduced more efficient and sustainable approaches, paving the way for broader industrial applications. Among them, the electrochemical reduction of CO<sub>2</sub> has become a particularly effective method of CO<sub>2</sub> conversion.<sup>5–9</sup>

Nevertheless, the high overpotential associated with the catalysts, as well as the inherent thermodynamic stability of CO<sub>2</sub> and the increased activation energy barrier, hinders the

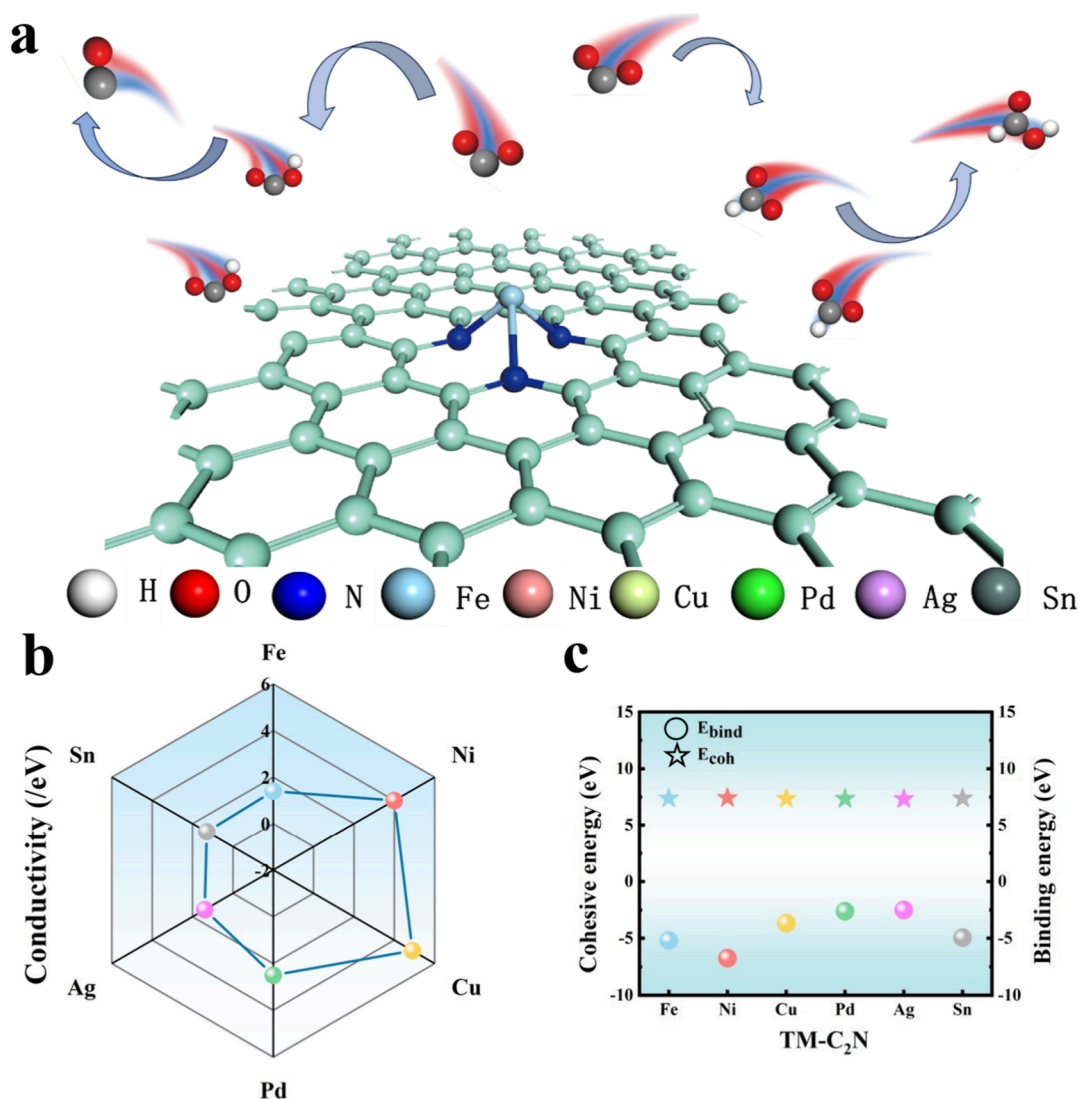
wider application of CO<sub>2</sub>RR.<sup>10–12</sup> Moreover, HER often supersedes CO<sub>2</sub>RR as a side reaction due to its relatively lower overpotential, further complicating the process. It is, therefore, imperative to design and develop efficient electrocatalysts that can diminish the overpotential required for CO<sub>2</sub> reduction, a pursuit of significant scientific and technological importance.<sup>13,14</sup>

Among various catalytic materials, single-atom catalysts (SACs) have attracted widespread attention due to their exceptional catalytic activity and remarkable atomic efficiency. SACs were first synthesized in 2011 and have demonstrated outstanding catalytic performance in the reduction of CO<sub>2</sub>.<sup>15,16</sup> Designing and synthesizing more efficient metal single-atom catalysts is both scientifically and practically significant. Substrate materials with large surface areas and excellent electronic and thermal properties, such as metal oxides, 2D materials, graphene, hexagonal boron nitride, and 2D polymer

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**Figure 1.** (a) Enhanced CO<sub>2</sub> conversion with tuned transition metal-loaded C<sub>2</sub>N. (b) Conductivity of different catalyst materials based on the height of the DOS curve at the Fermi level. (c) Cohesive energy and binding energy of TM-C<sub>2</sub>N structures.

metal phthalocyanine sheets, have been employed as catalysts carriers.<sup>17–19</sup> Recently, a novel two-dimensional material C<sub>2</sub>N monolayer with high structural and thermal stability has been discovered. The special structural composition can provide the best metal impurity site and significantly improve the catalysts performance.<sup>20–23</sup>

A deep understanding of the underlying mechanisms governing the electrocatalytic reduction of CO<sub>2</sub> is critical for the rational design and optimization of catalysts. Extensive research has been conducted to explore the potential mechanisms of CO<sub>2</sub>RR, with a specific focus on the use of density functional theory (DFT) calculations to provide a robust quantum chemical framework.<sup>24–26</sup> Advances in nanotechnology have enabled the fine-tuning of catalysts structures at the nanoscale, thereby enhancing stability, catalytic activity, and product selectivity.<sup>27,28</sup> Notably, SACs, double-atom catalysts (DACs), and triple-atom catalysts (TACs), characterized by their highly dispersed active sites, have demonstrated low overpotentials, elevated conversion efficiencies, and commendable product selectivity, along with an inherent capacity to suppress the hydrogen evolution

reaction, which augurs well for their broad application prospects.<sup>29–32</sup>

Prior research has established that SACs and DACs, when modified with metal atoms such as Cu, Pd, Ag, and Pt, exhibit superior CO<sub>2</sub>RR catalytic activity.<sup>33–37</sup> Research has explored the use of copper-doped nickel metal as a catalytic substrate for the CO<sub>2</sub> reduction reaction.<sup>38</sup> Building on this, a nitrogen/sulfur-coordinated iron–nickel dual-atom catalyst was developed.<sup>39</sup> Separate studies have shown that “vacancies” and nitrogen doping in graphene can effectively stabilize iron atom catalysts.<sup>40</sup> Concurrently, a single-atom tin catalyst based on SnS<sub>2</sub> material was found to efficiently convert CO<sub>2</sub> to formic acid.<sup>41</sup> Additionally, research indicates that incorporating hydrogen into copper single-atom catalysts can modulate their microstructure.<sup>42</sup> While single-metal sites have been confirmed to possess significant potential for CO<sub>2</sub> conversion, most existing research has been limited to individual metals and specific products. In contrast, this study systematically compares the performance of a series of single-atom catalysts in the electrochemical CO<sub>2</sub> reduction reaction, aiming to reveal the decisive role of the intrinsic properties of metals in determining catalytic pathways. Selecting appropriate catalysts

materials can effectively curtail the aggregation of transition metal single atoms, simplifying the synthesis process and bolstering catalyst stability through augmented interfacial interactions.<sup>43</sup> Nitrogen-doped graphene ( $C_2N$ ), widely utilized as a support material for SACs, holds significant application potential owing to its periodic uniformity and elevated thermal stability.<sup>44–48</sup> However, given the exorbitant costs and scarcity associated with noble metals, there is an imperative to develop cost-effective, high-performance catalysts as viable alternatives.

In this study, the present investigation systematically assesses the catalytic efficacy of metal single-atom materials anchored on nitrogen-doped graphene (denoted as TM- $C_2N$ , with TM representing Fe, Ni, Cu, Pd, Ag, and Sn), employing first-principles computational methodologies. Initially, the stability and electronic configuration of the TM- $C_2N$  catalysts are ascertained through cohesive energy calculations. Subsequently, leveraging d-band center theory and a comprehensive electronic structure analysis, the adsorption energy and activation propensity of  $CO_2$  are elucidated. Ultimately, through an analysis of the Gibbs free energy profiles, the optimal reaction pathways for the formation of various products, alongside their corresponding limiting and overpotentials, are delineated, thereby shedding light on the underlying mechanisms governing the  $CO_2$  reduction process.

## EXPERIMENTAL SECTION

The computational work was performed using the Dmol<sup>3</sup> software, which is based on density functional theory under periodic boundary conditions.<sup>49,50</sup> The elements involved include carbon, nitrogen, oxygen, hydrogen, copper, iron, palladium, tin, silver, and nickel, with their valence electron configurations being  $2s^22p^2$ ,  $2s^22p^3$ ,  $2s^22p^4$ ,  $1s^1$ ,  $3d^{10}4s^1$ ,  $3d^64s^2$ ,  $4d^{10}$ ,  $5s^25p^2$ ,  $4d^{10}5s^1$ , and  $3d^84s^2$ , respectively. The interaction between valence and core electrons was described using the projector augmented wave (PAW) method. The exchange-correlation effects were simulated using the Perdew–Burke–Ernzerhof (PBE) functional within the generalized gradient approximation (GGA) framework.<sup>51</sup> To improve the precision of the total energy calculations, spin polarization was incorporated, and a dispersion correction (Grimme D2) was implemented.<sup>52,53</sup> For structural optimization and electronic property calculations, strict convergence criteria were adopted: The convergence threshold for total energy was set to  $10^{-5}$  eV, and the convergence threshold for atomic forces was 0.01 eV/Å. A  $\Gamma$ -point-centered  $k$ -point mesh with a  $3 \times 3 \times 1$  grid was employed for Brillouin zone sampling, which balances computational efficiency and accuracy for 2D monolayer systems by sufficiently covering the electronic states near the Fermi level. To eliminate spurious interactions between adjacent periodic supercells, a vacuum layer of 20 Å was introduced vertically to the TM- $C_2N$  monolayer plane; this thickness is greater than the typical van der Waals interaction range ( $\approx 5$ – $10$  Å), ensuring no artificial coupling between periodic images. An implicit solvation model was employed to represent the solvent, with the dielectric constant set to 78.4 for water.<sup>54</sup>

## RESULTS AND DISCUSSION

### Stability of TM- $C_2N$ Catalysts

The single-atom catalyst model is constructed on a  $2 \times 2$  supercell of the  $C_2N$  lattice. The metal single atoms are

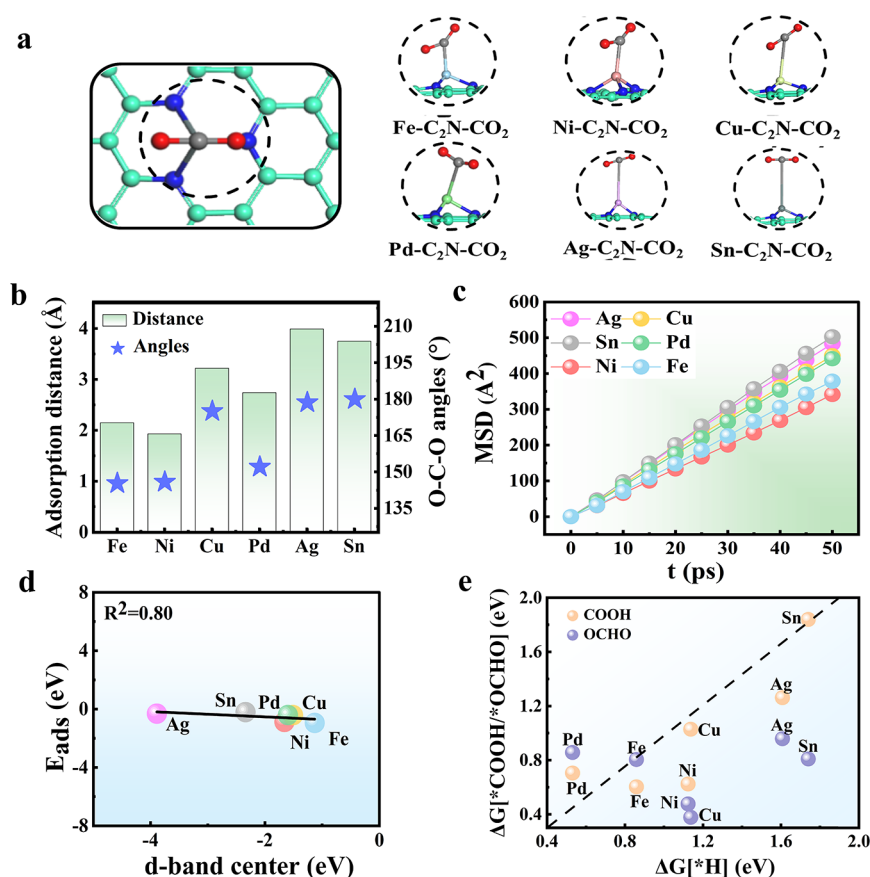
positioned atop the  $C_2N$  lattice, each metal atom covalently bonded to three nitrogen atoms (Figure S1). To elucidate the conductivity and stable configuration of the  $C_2N$  catalysts for  $CO_2$ RR modified by six distinct metals, a schematic representation is provided in Figure 1a. The catalysts can be used as a good transfer material, and its electrical conductivity is one of the most important indicators. The density of states (DOS) curve height at the Fermi level for TM- $C_2N$  was calculated, serving as a pivotal parameter for assessing catalyst conductivity.<sup>55–59</sup> In Figure 1b, the electrical conductivity of all metal-doped monolayers is enhanced, which indicates that the method of metal atomic modification is suitable for improving the performance of the catalysts. It is worth noting that the increase in conductivity of the monolayer materials doped with Cu and Ni atoms is the most significant.

Given that stability is a fundamental requirement for catalyst applications, the stability of the TM- $C_2N$  catalysts is gauged through cohesive energy calculations and the binding energies of the metal atoms.<sup>60–62</sup> The detailed calculation of cohesive energy can be found in the Supporting Information. As shown in Figure 1c, the cohesive energies of the TM- $C_2N$  catalysts are ranked in the order of Fe- $C_2N$ , Ni- $C_2N$ , Cu- $C_2N$ , Pd- $C_2N$ , Ag- $C_2N$ , and Sn- $C_2N$ , with Ni- $C_2N$  exhibiting the highest stability. The cohesive energies of the TM- $C_2N$  catalysts surpass those of carbides (4.12–6.45 eV/atom)<sup>63</sup> and silicene (5.16 eV/atom),<sup>64</sup> underscoring their superior stability. To assess the thermodynamic stability of TM- $C_2N$  structures, we performed ab initio molecular dynamics (AIMD) simulations. The simulations were conducted at 500 K for 10 ps, with structural snapshots recorded every 1 fs. As illustrated in Figures S2 and S3, the results demonstrate that TM- $C_2N$  materials maintain structural integrity under these conditions, highlighting their inherent stability.

Furthermore, leveraging the formation environment of single atoms on the nitrogen–carbon carrier, the binding energies at the interface between the carrier site and metal atoms are harnessed to evaluate the stability of the single-atom catalysts. The results indicate that these binding energies possess relatively high magnitudes, thereby ensuring the stability of the catalysts. To assess the stability of electrocatalysts during long-term operation under reaction conditions, the dissolution potential is computed.

To enhance our comprehension of the interactions between metal single atoms and  $C_2N$ , computational analyses of DOS, PDOS, and charge density at the interface for TM- $C_2N$  catalysts were conducted. As depicted in Figures S5–S7, in the vicinity of the Fermi level, a pronounced overlap of electronic wave function is observed between the d orbitals of the metal single atoms and the p orbitals of nitrogen within the  $C_2N$  lattice, signifying interactions at the atomic level. Notably, in the cases of Cu- $C_2N$  and Ni- $C_2N$ , an increased number of overlapping peaks are evident, suggesting a more pronounced interaction between the Cu and Ni atoms and the nitrogen atoms when compared to other metals studied. This observation is instrumental in explaining the enhanced stability of the Cu- $C_2N$  and Ni- $C_2N$  catalysts. Furthermore, the charge density analysis discloses considerable electron accumulation and delocalization at the interface of metal atoms and the substrate, substantiating the existence of interactions. These interactions culminate in augmented bonding energies and diminished bond lengths between the Cu and Ni atoms and the nitrogen atoms.





**Figure 2.** (a) Optimal structures of CO<sub>2</sub> on the TM-C<sub>2</sub>N catalysts. (b) Activated C–M bond length and O–C–O bond angle bending. (c) Diffusion behaviors of CO<sub>2</sub> on TM-C<sub>2</sub>N. (d) Linear relationship between the d-band center of TM-C<sub>2</sub>N and the adsorption energy of CO<sub>2</sub>. (e) Free energy change of the first hydrogenation steps in CO<sub>2</sub>RR vs HER.

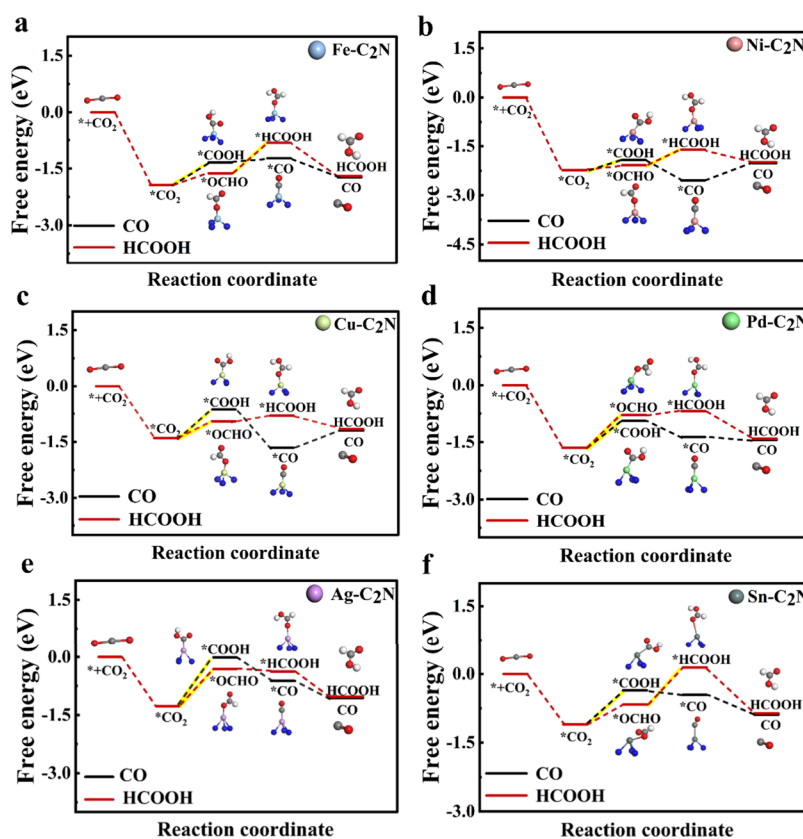
### Activation and Selectivity

Stable CO<sub>2</sub> adsorption is a key prerequisite for the efficient operation of CO<sub>2</sub>RR. The extent of CO<sub>2</sub> activation is indicative of the catalysts' performance to a certain degree. As depicted in Figure 2a, CO<sub>2</sub> molecules stably adsorb onto the catalyst surface, with significant alterations in molecular configuration, signifying effective initial activation (Figure S8). The results indicate in Figure 2b that after the adsorption of CO<sub>2</sub> on the surface, there is a significant change in the C–M bond length (when CO<sub>2</sub> is bonded to the surface) within the range from 1.88 to 3.50 Å. Meanwhile, the O–C–O bond angle also shows a bending from 145.3 to 179.4°. These remarkable changes in the structural parameters fully confirm the effective activation of the CO<sub>2</sub> molecule. Additionally, our detailed analysis of the DOS curves between gas molecules and catalyst surface active sites reveals significant interactions for Cu and Ni with CO<sub>2</sub>, generating distinct new energy peaks (Figures S9 and S10). Notably, the hybridization between metal d orbitals and nonmetal p orbitals is the predominant factor.

The diffusion characteristics of CO<sub>2</sub> on the catalyst surface are determinants of the reaction rate. To assess the practical application performance of these catalysts, molecular dynamics simulations were employed to calculate the diffusion coefficients of CO<sub>2</sub> across various material surfaces, thereby characterizing their diffusion properties. To study the diffusion behavior of CO<sub>2</sub> on the TM-C<sub>2</sub>N surface, we performed molecular dynamics (MD) simulations. The detailed calculation can be found in the Supporting Information and Figures

S11 and S12. Ni-C<sub>2</sub>N exhibits excellent CO<sub>2</sub> adsorption capacity, which is attributed to its low diffusion kinetic characteristics. Meanwhile, CO<sub>2</sub> molecules do not exhibit stable adsorption on the uppermost surface and show a preferential distribution throughout the bulk phase. Initially, the MSD for the TM-C<sub>2</sub>N catalysts was calculated and is presented in Figure 2c. The results indicate that Ni-C<sub>2</sub>N has good diffusion performance, as inferred from the slope of the MSD plot. Furthermore, the diffusion coefficient, which impacts the surface diffusion of CO<sub>2</sub>, can be derived from Einstein's equation for diffusion. The diffusion coefficients of CO<sub>2</sub> are 1.66 for Fe-C<sub>2</sub>N, 1.16 for Ni-C<sub>2</sub>N, 1.46 for Cu-C<sub>2</sub>N, 1.47 for Pd-C<sub>2</sub>N, 1.56 for Ag-C<sub>2</sub>N, and 1.28 for Sn-C<sub>2</sub>N. The magnitude of the diffusion coefficient directly influences CO<sub>2</sub> surface diffusion. More stable CO<sub>2</sub> adsorption and enhanced catalytic efficacy are correlated with a smaller coefficient.

To further elucidate the adsorption behavior of CO<sub>2</sub> on the catalysts, calculations of the d-band center for the metal single-atom catalysts were performed. Adsorption energy is specified as  $E_{\text{ads}} = E_{\text{total}} - (E_{\text{monolayer}} + E_{\text{gas}})$ , where  $E_{\text{ads}}$  represents the adsorption energy (eV),  $E_{\text{total}}$  is the total energy,  $E_{\text{monolayer}}$  is the energy of the two-dimensional monolayer, and  $E_{\text{gas}}$  is the energy of the gas (eV). Figure 2d delineates the correlation between the d-band center of the metal single atoms and the CO<sub>2</sub> adsorption energy. The d-band center shows a significant correlation with catalytic activity ( $R^2 = 0.80$ ). This aligns with the predictive trend of the d-band center theory, which states that a moderate d-band center enables the formation of an optimal binding energy between the catalysts and reactants,



**Figure 3.** Gibbs free energy distribution diagram of the  $\text{CO}_2$  reduction pathway. (a) Fe- $\text{C}_2\text{N}$ , (b) Ni- $\text{C}_2\text{N}$ , (c) Cu- $\text{C}_2\text{N}$ , (d) Pd- $\text{C}_2\text{N}$ , (e) Ag- $\text{C}_2\text{N}$ , and (f) Sn- $\text{C}_2\text{N}$ .

thereby enhancing catalytic efficiency. From an electronic structure perspective, such moderate binding energy typically corresponds to lower electron occupancy in the reactant-catalyst antibonding orbitals, avoiding the weakening of binding caused by excessive occupation of antibonding states. From a kinetic perspective, the propensity for the HER to occur as a side process can significantly hinder the advancement of the electrocatalytic reduction of  $\text{CO}_2$ . Consequently, it is imperative for efficient catalysts to effectively suppress the HER. The Gibbs free energies of hydrogen adsorption on the surfaces of Fe- $\text{C}_2\text{N}$ , Ni- $\text{C}_2\text{N}$ , Cu- $\text{C}_2\text{N}$ , Pd- $\text{C}_2\text{N}$ , Ag- $\text{C}_2\text{N}$ , and Sn- $\text{C}_2\text{N}$ , catalysts was determined to be 0.85, 1.12, 1.13, 0.53, 1.60, and 1.74 eV, respectively. When compared with the Gibbs free energy of  $\text{CO}_2$  adsorption (Table S1),  $\text{CO}_2$  adsorption is energetically more favorable than hydrogen adsorption in this reaction system. These findings offer reliable data and a robust theoretical foundation for elucidating the suppression of the HER.

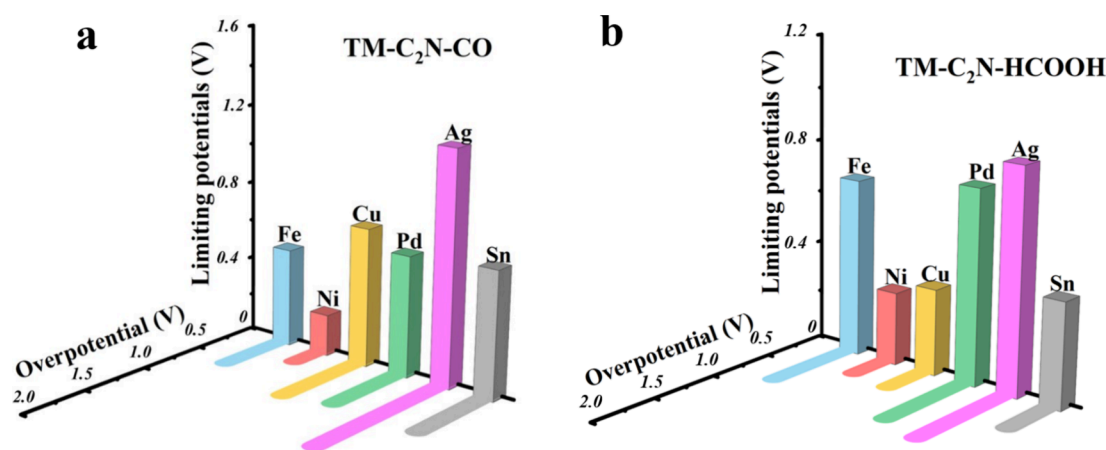
The results further clarify the intrinsic reason for the effective suppression of the HER by TM- $\text{C}_2\text{N}$ , the synergistic effect of thermodynamic preference and electronic structure regulation. Thermodynamically, the positive Gibbs free energy of  $\text{H}^+$  adsorption stands in stark contrast to the negative adsorption energy of  $\text{CO}_2$ , enabling  $\text{CO}_2$  to competitively occupy the active TM sites and reduce the chance of  $\text{H}^+$  binding, which directly hinders the initial step of HER. Electronically, charge density and PDOS analyses confirm that the d orbitals of TM atoms form strong hybridization with the p orbitals of N atoms in  $\text{C}_2\text{N}$ . This orbital interaction redirects the electron density of active sites to interact with the O 2p orbitals of  $\text{CO}_2$  rather than the 1s orbital of  $\text{H}^+$ , thereby

weakening the binding strength between H and the catalyst surface.

Furthermore, based on the principles of adsorption energetics, it is apparent that  $\text{CO}_2$  is more readily adsorbed onto the TM- $\text{C}_2\text{N}$  catalyst surface compared to hydrogen, which effectively mitigates the onset of the HER. Additionally, during the initial protonation step of  $\text{CO}_2$  hydrogenation, two key intermediates are produced:  $^*\text{COOH}$  (via the reaction  $^*\text{CO}_2 + \text{H}^+ + \text{e}^- \rightarrow ^*\text{COOH}$ ) and  $^*\text{OCHO}$  (via the reaction  $^*\text{CO}_2 + \text{H}^+ + \text{e}^- \rightarrow ^*\text{OCHO}$ ). Meanwhile, for the HER, the formation of the  $^*\text{H}$  intermediate (via the reaction  $^* + \text{H}^+ + \text{e}^- \rightarrow ^*\text{H}$ ) is observed. As depicted in Figure 2e, when  $\Delta G(^*\text{COOH})$  and  $\Delta G(^*\text{OCHO})$  are less than  $\Delta G(^*\text{H})$ , the reaction requires less energy and is more conducive to product formation, upon comparison of the Gibbs free energy changes  $\Delta G(^*\text{COOH})$  and  $\Delta G(^*\text{OCHO})$  with  $\Delta G(^*\text{H})$ , the TM- $\text{C}_2\text{N}$  catalysts are situated in the lower right region of the selectivity diagram, which is the  $\text{CO}_2\text{RR}$  selectivity advantage zone. A preference for the formation of  $^*\text{COOH}$  or  $^*\text{OCHO}$  intermediates is indicated by this structural positioning. Taking these findings into account, it is evident that TM- $\text{C}_2\text{N}$  catalysts demonstrate a marked selectivity for  $\text{CO}_2\text{RR}$ , concurrently efficaciously suppressing the HER as a competing side reaction.

### $\text{CO}_2$ Reduction Pathway

Figure 3 presents the adsorption configurations and the corresponding free energy landscapes for the intermediates produced during the  $\text{CO}_2$  reduction process on TM- $\text{C}_2\text{N}$  catalysts.  $\text{CO}_2$  adsorption happens on its own without needing extra energy. A negative Gibbs free energy also gives an extra push from a thermodynamic point of view for the  $\text{CO}_2$  reduction reaction to take place. This negative Gibbs free



**Figure 4.** Limiting potential and overpotential of CO<sub>2</sub>RR. (a) CO and (b) HCOOH.

energy helps key parts like \*COOH and \*CO form more easily after CO<sub>2</sub> gets adsorbed. As a result, the energy needed to start the reaction goes down, and the reaction speeds up. HCOOH is predominantly produced via the \*OCHO intermediate, a pathway that entails the direct addition of H to CO<sub>2</sub>, culminating in the formation of HCOOH. CO is primarily generated via the \*COOH intermediate, which subsequently leads to the release of CO. The CO<sub>2</sub>-to-CO conversion pathway on TM-C<sub>2</sub>N catalysts is delineated as follows: CO<sub>2</sub> → \*CO<sub>2</sub> → \*COOH → \*CO → CO, with the rate-determining step being the transformation of \*CO<sub>2</sub> to \*COOH. The maximum Δ*G* for the rate-limiting step are 0.61 eV for Fe-C<sub>2</sub>N, 0.30 eV for Ni-C<sub>2</sub>N, 0.80 eV for Cu-C<sub>2</sub>N, 0.71 eV for Pd-C<sub>2</sub>N, 1.27 eV for Ag-C<sub>2</sub>N, and 0.74 eV for Sn-C<sub>2</sub>N, respectively.

The pathway for CO<sub>2</sub> conversion to HCOOH on TM-C<sub>2</sub>N catalysts is as follows: CO<sub>2</sub> → \*OCHO → \*HCOOH → HCOOH, with the rate-determining step identified as the initial \*CO<sub>2</sub> to \*OCHO conversion. Among them, the rate-determining step for Fe, Ni, and Sn is \*OCHO to \*HCOOH. The corresponding maximum Δ*G* changes for the rate-limiting step in the HCOOH formation pathway are 0.80 eV for Fe-C<sub>2</sub>N, 0.40 eV for Ni-C<sub>2</sub>N, 0.45 eV for Cu-C<sub>2</sub>N, 0.86 eV for Pd-C<sub>2</sub>N, 0.97 eV for Ag-C<sub>2</sub>N, and 0.52 eV for Sn-C<sub>2</sub>N. Among them, Ni shows the best advantages in the CO conversion path. Adsorption energy calculations for the \*COOH and \*OCHO intermediates on Ni-C<sub>2</sub>N yield values of −2.44 and −2.35 eV, respectively. These results highlight the significant superiority of Ni-C<sub>2</sub>N in activating intermediates, leading to a reduced free energy barrier for the reduction pathway.

### Catalytic Performance

To quantitatively delineate catalyst performance, the limiting potential (*U*<sub>L</sub>) and overpotential (*η*) were meticulously calculated and are hereby utilized as metrics of catalytic efficacy. Figure 4a presents the limiting potentials and overpotentials associated with the conversion of CO<sub>2</sub> to CO across a spectrum of catalysts, including Fe-C<sub>2</sub>N, Ni-C<sub>2</sub>N, Cu-C<sub>2</sub>N, Pd-C<sub>2</sub>N, Ag-C<sub>2</sub>N, and Sn-C<sub>2</sub>N. Overpotential, which is the difference between the limiting potential for product formation and the equilibrium potential, is a key indicator for assessing the performance of electrochemical systems. The theoretical equilibrium potentials for the reduction reactions yielding CO and HCOOH are established at 0.21 and 0.24 V, respectively. A near-zero *U*<sub>L</sub>, as observed in Figure 4a, signifies a lower energetic barrier to produce the corresponding

product. The *U*<sub>L</sub> for CO production is ranked in ascending order as follows: Ni-C<sub>2</sub>N, Fe-C<sub>2</sub>N, Pd-C<sub>2</sub>N, Cu-C<sub>2</sub>N, Sn-C<sub>2</sub>N, and Ag-C<sub>2</sub>N. Similarly, the overpotentials required for CO generation are ordered from the lowest to the highest: Ni-C<sub>2</sub>N, Fe-C<sub>2</sub>N, Pd-C<sub>2</sub>N, Cu-C<sub>2</sub>N, Sn-C<sub>2</sub>N, and Ag-C<sub>2</sub>N. Considering both the limiting potential and overpotential, Ni-C<sub>2</sub>N exhibits the best performance in CO production (Figure 4b). Consequently, the utilization of TM-C<sub>2</sub>N catalysts in the context of CO<sub>2</sub>RR is deemed viable. This study has thus explored the exceptional performance of TM-C<sub>2</sub>N catalysts in the electrocatalytic reduction of CO<sub>2</sub>, with Ni-C<sub>2</sub>N surpassing the performance of the other five catalysts evaluated.

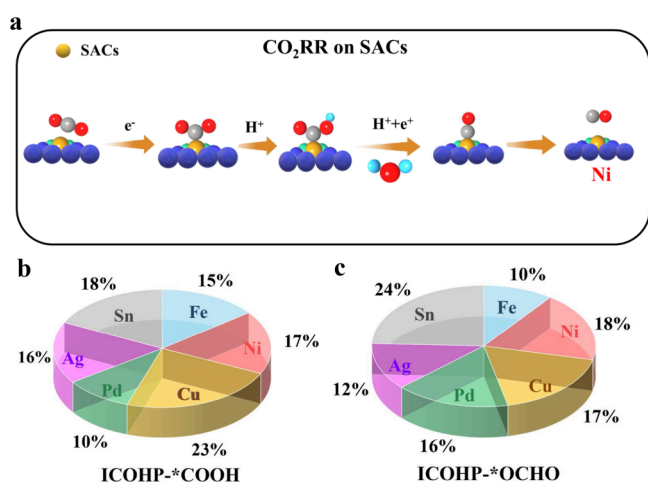
### Catalytic Mechanism

In this section, the molecular orbital framework is explored to unravel the mechanisms underpinning the superior performance of Ni-C<sub>2</sub>N catalysts, with a specific focus on the interactions between key intermediates and catalytically active sites. The role of bonding orbitals is critical in adsorption processes, where the initial interaction must be carefully balanced—neither too strong nor too weak. Therefore, crystal orbital Hamilton population (COHP) calculations were performed to analyze the key intermediates associated with the most effective transition metal catalysts (Figure 5a). Achieving an optimal balance in these interactions is essential for developing high-performance catalysts.

Figures S13 and S14 display the optimal adsorption geometries, charge density distributions, and DOS for \*COOH and \*OCHO on TM-C<sub>2</sub>N catalysts. Notable charge redistribution is observed at the interaction sites between \*COOH/\*OCHO and TM-C<sub>2</sub>N. The DOS analysis indicates significant orbital hybridization between nonmetal p orbitals and metal d orbitals, which enhances the desorption and reduction of CO and HCOOH on TM-C<sub>2</sub>N catalysts. Therefore, these findings underscore the catalytic performance of TM-C<sub>2</sub>N for the conversion of CO<sub>2</sub> to CO and HCOOH.

Figure 5b,c illustrates the bonding and antibonding regions at the Fermi level between the TM-C<sub>2</sub>N surface and gas-phase atoms, with the corresponding ICOHP (Integrated Crystal Orbital Hamiltonian Population) values provided for clarity. The ICOHP was calculated to quantify the bond strength, with more negative values indicating stronger bonding. Detailed calculations are presented in Figure S15. A higher ICOHP value corresponds to stronger bond energy and greater energy required for desorption. The adsorption strength between a catalyst and reactant molecules should be maintained within an





**Figure 5.** (a) Intermediates of \*COOH and \*OCHO at CO<sub>2</sub>RR. COHP analysis for different catalysts regarding intermediates of (b) \*COOH and (c) \*OCHO.

optimal range. Excessively strong interactions can lead to irreversible occupation of active sites, impeding the reaction pathway, while overly weak interactions are insufficient to activate the reactant molecules, making it difficult to overcome the reaction energy barrier. Thus, a moderate ICOHP value is necessary to achieve optimal adsorption.

For CO production (Figure 5b), the ICOHP values for \*COOH intermediates on Fe, Ni, Cu, Pd, Ag, and Sn were 1.10, 1.30, 1.78, 0.77, 1.27, and 1.33 eV, respectively. To more intuitively compare the relative bonding strengths of different TM-C<sub>2</sub>N catalysts toward key intermediates, we calculated the ICOHP percentage. This parameter is obtained by dividing the ICOHP value of a given catalyst by the sum of the ICOHP values of all catalysts and then multiplying by 100%. The resulting percentage directly reflects the relative contribution and ranking of each catalyst's bonding strength within the entire set. Ni-C<sub>2</sub>N corresponds to a percentage of 17%, representing that the contribution of its bonding orbitals with \*COOH accounts for 17% of the total COHP value. It is in a moderately proportionate position. These findings support the idea that intermediates such as \*COOH and SACs significantly enhance catalytic activity.

For HCOOH (Figure 5c), the ICOHP values for \*OCHO intermediates on Fe, Ni, Cu, Pd, Ag, and Sn were 0.17, 0.28, 0.27, 0.25, 0.19, and 0.37 eV, respectively. For the CO pathway, the key intermediate is \*COOH. Calculations indicate that both the adsorption energy and ICOHP percentage of \*COOH on Ni-C<sub>2</sub>N exhibit moderate strength. This optimal bonding ensures stable formation of \*COOH while preventing excessive stabilization that would hinder subsequent \*CO desorption, thereby enabling the CO pathway to achieve the lowest limiting potential on Ni-C<sub>2</sub>N. In contrast, Cu-C<sub>2</sub>N exhibits excessively strong bonding with \*COOH, which may impede its further transformation or complicate \*CO desorption, consequently increasing the overpotential of the CO pathway. For the HCOOH pathway, the key intermediate is \*OCHO. Notably, although the bonding strength between Ni-C<sub>2</sub>N and \*OCHO is not the strongest, it establishes a favorable balance with its bonding strength to \*COOH. This "moderate-to-weak" bonding characteristic toward both \*COOH and \*OCHO renders

Ni-C<sub>2</sub>N thermodynamically more selective for CO production, as CO desorption is kinetically more favorable.

Compared to other catalysts, the adsorption energies of \*COOH and \*OCHO demonstrate moderate interactions with Ni active sites, respectively. This optimal interaction ensures the superior performance of Ni-C<sub>2</sub>N in CO production synthesis. The findings demonstrate that single-atom interactions in SACs effectively modulate the electronic structure, optimizing active sites for the dynamic adsorption of key reaction intermediates. Notably, SACs display moderate adsorption energies for intermediates such as \*COOH and \*OCHO, pivotal for their remarkable catalytic activity. The study concludes that Ni-C<sub>2</sub>N is the most effective catalysts for both carbon monoxide production.

## CONCLUSIONS

In this study, DFT calculations were employed to design TM-C<sub>2</sub>N catalysts and evaluate their performance in the electrochemical reduction of CO<sub>2</sub> to CO and HCOOH. The cohesive energies of these catalysts, spanning 7.30–7.36 eV/atom, underscore their stability. The strong orbital overlap between the metal atoms and the C<sub>2</sub>N substrate further reinforces the structural integrity of the TM-C<sub>2</sub>N complex. CO<sub>2</sub> adsorption studies reveal that TM-C<sub>2</sub>N facilitates CO<sub>2</sub> capture and activation, driven by the interplay between surface architecture and active TM sites. Adsorption energies for CO<sub>2</sub> range from −0.94 to −3.81 eV, with notable changes in O–C–O bond angles and C–O bond lengths, indicating effective activation. Gibbs free energy calculations show that the Ni-C<sub>2</sub>N pair exhibits lower energy barriers for the \*COOH and \*HCOOH intermediates, while significantly inhibiting the HER. The limiting potentials of these catalysts indicate their excellent catalytic activity. Among them, Ni-C<sub>2</sub>N is favorable for the rate-determining steps of \*CO<sub>2</sub>→\*COOH and \*OCHO→\*HCOOH, with *U*<sub>L</sub> values of 0.30 and 0.40 V respectively, which demonstrates that Ni-C<sub>2</sub>N has better performance in the CO reduction pathway. In addition, the COHP values between the key intermediates of CO<sub>2</sub> reduction and the catalyst active site reveal that moderate adsorption ensures high catalytic performance. These results underscore the potential of transition metals as cost-effective alternatives to precious metals for efficient CO<sub>2</sub> electrocatalysis.

## ASSOCIATED CONTENT

### Data Availability Statement

Data will be made available on request.

### Supporting Information

The Supporting Information is available free of charge at <https://pubs.acs.org/doi/10.1021/acs.jpcc.5c07261>.

Computational details including atomic structure models, ab initio molecular dynamics simulation results, density of states and charge density analysis, CO<sub>2</sub> adsorption configurations, molecular dynamics simulation for diffusion, Gibbs free energy data, and crystal orbital Hamilton population analysis for key intermediates (PDF)

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## Author Contributions

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## Notes

The authors declare no competing financial interest.

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